#### PAPER

## A Josephson junction with *h*-BN tunnel barrier: observation of low critical current noise

To cite this article: Jifa Tian et al 2021 J. Phys.: Condens. Matter 33 495301

View the article online for updates and enhancements.

#### You may also like

- Direct observation of layer-stacking and oriented wrinkles in multilayer hexagonal boron nitride
   Lingxiu Chen, Kenan Elibol, Haifang Cai et al.
- <u>Non-chemical fluorination of hexagonal</u> boron nitride by high-energy ion irradiation Shiro Entani, Konstantin V Larionov, Zakhar I Popov et al.
- Dependence of the polycarbonate mechanical performances on boron nitride flakes morphology Emanuele Lago, Peter S Toth, Silvia Gentiluomo et al.



## IOP ebooks<sup>™</sup>

Bringing together innovative digital publishing with leading authors from the global scientific community.

Start exploring the collection-download the first chapter of every title for free.

J. Phys.: Condens. Matter 33 (2021) 495301 (6pp)

# A Josephson junction with *h*-BN tunnel barrier: observation of low critical current noise

### Jifa Tian<sup>1,2,3,\*</sup>, Luis A Jauregui<sup>1,4</sup>, C D Wilen<sup>5</sup>, Albert F Rigosi<sup>3</sup>, David B Newell<sup>3</sup>, R McDermott<sup>5</sup> and Yong P Chen<sup>1,6,7,8,9,\*</sup>

<sup>1</sup> Department of Physics and Astronomy and Birck Nanotechnology Center, Purdue University, West Lafayette, IN 47907, United States of America

<sup>2</sup> Department of Physics and Astronomy, University of Wyoming, Laramie, WY 82071, United States of America

<sup>3</sup> National Institute of Standards and Technology, Gaithersburg, MD 20899, United States of America
 <sup>4</sup> Department of Physics and Astronomy, University of California, Irvine, CA 92697, United States of America

<sup>5</sup> Department of Physics, University of Wisconsin-Madison, Madison, WI 53706, United States of America

<sup>6</sup> Purdue Quantum Science and Engineering Institute, Purdue University, West Lafayette, IN 47907, United States of America

<sup>7</sup> School of Electrical and Computer Engineering, Purdue University, West Lafayette, IN 47907, United States of America

<sup>8</sup> Institute of Physics and Astronomy and Villum Centers for Dirac Materials and for Hybrid Quantum Materials, Aarhus University, 8000 Aarhus-C, Denmark

<sup>9</sup> WPI-AIMR International Research Center for Materials Sciences, Tohoku University, Sendai 980-8577, Japan

E-mail: jtian@uwyo.edu and yongchen@purdue.edu

Received 22 July 2021, revised 30 August 2021 Accepted for publication 14 September 2021 Published 1 October 2021



#### Abstract

Decoherence in quantum bits (qubits) is a major challenge for realizing scalable quantum computing. One of the primary causes of decoherence in qubits and quantum circuits based on superconducting Josephson junctions is the critical current fluctuation. Many efforts have been devoted to suppressing the critical current fluctuation in Josephson junctions. Nonetheless, the efforts have been hindered by the defect-induced trapping states in oxide-based tunnel barriers and the interfaces with superconductors in the traditional Josephson junctions. Motivated by this, along with the recent demonstration of 2D insulator *h*-BN with exceptional crystallinity and low defect density, we fabricated a vertical NbSe<sub>2</sub>/*h*-BN/Nb Josephson junction consisting of a bottom NbSe<sub>2</sub> superconductor thin layer and a top Nb superconductor spaced by an atomically thin *h*-BN layer. We further characterized the superconducting current and voltage (I-V) relationships and Fraunhofer pattern of the NbSe<sub>2</sub>/*h*-BN/Nb junction. Notably, we demonstrated the critical current noise (1/f noise power) in the *h*-BN-based Josephson device is at least a factor of four lower than that of the previously studied aluminum oxide-based Josephson junctions. Our work offers a strong promise of *h*-BN as a novel tunnel barrier for high-quality Josephson junctions and qubit applications.

Keywords: Josephson junctions, boron nitride, critical current noise

(Some figures may appear in colour only in the online journal)

<sup>\*</sup> Authors to whom any correspondence should be addressed.

#### 1. Introduction

Josephson junctions underlie the operations of many superconductor-based device applications ranging superconducting quantum from interference devices (SQUIDs) to superconducting quantum bits (qubits) for quantum computing [1, 2]. A representative superconductor-insulator-superconductor (S/I/S) Josephson junction normally consists of two superconductors separated by a thin insulator as a tunnel barrier. Typically, the tunnel barriers in the S/I/S junctions are made of a thin metal oxide layer [2], such as aluminum oxide  $(AlO_x)$  [3]. One of the grand challenges of realizing scalable quantum computing using such a mesoscopic S/I/S junction device is the minimization of decoherence of qubits [4, 5]. It has been shown that the coherence time of qubits in S/I/S Josephson junctions is largely affected by the quality of the individual materials [6-9], particularly, the insulating spacer as well as the interfaces between the insulating spacer and its adjacent superconductors, where charge trapping states can inevitably exist. For example, one of the serious problems is the critical-current fluctuation caused by the charge trapping at the defect sites in the S/I/S Josephson junctions [4, 5, 10-12]. In recent years, a long-standing goal in this field is to create a high-quality and defect-free insulating spacer that may lead to improved properties of the superconducting Josephson junctions.

Ever since the mechanical exfoliation of atomically thin graphene layers from bulk graphite [13, 14] and their deposition or dry-transferring on various substrates [14], significant progress has also been made in other two-dimensional (2D) materials, including examples such as insulating hexagonal boron nitride (h-BN) [15, 16], superconducting niobium diselenide (NbSe<sub>2</sub>) [6, 17], semiconducting molybdenum disulfide  $(MoS_2)$  [18, 19] and black phosphorene (BP) [20], etc. An attractive merit of using 2D insulators and semiconductors as tunnel barriers in high-quality Josephson junctions is that such 2D materials can be highly crystalline, atomically thin, and nearly defect-free. Recently, notable progress has been made in making designable Josephson junctions using insulating/semiconducting 2D materials. For instance, the Josephson effect in Al/MoS<sub>2</sub>/Al tunnel junctions using a MoS<sub>2</sub> thin layer as the tunnel barrier has been recently demonstrated [19]. People have developed an in situ technique to fabricate Nb/BP/Nb Josephson junctions and demonstrated good interfacial properties between Nb and BP by the Josephson effect [21]. Recently, 2D magnetic insulators (MIs) have also been used to fabricate high-quality magnetic Josephson junctions consisting of 2D superconductor/2D MI/2D superconductor van der Waals (vdW) heterostructures [22-24]. On the other hand, *h*-BN as the first discovered 2D insulator with a large band gap has attracted extensive attention [16, 25, 26], promising an ideal candidate for making single crystalline and defect-free insulating spacers for Josephson junctions. However, quantitatively analyzing the quality of the Josephson junction devices with an atomically thin h-BN layer is still missing. Specifically, how the 2D insulator tunnel barrier affects the critical current noise in the Josephson junctions remains unanswered. In the

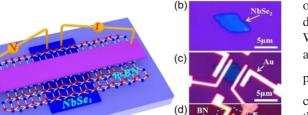
present study, we have fabricated a vertical NbSe<sub>2</sub>/*h*-BN/Nb Josephson junction (figure 1(a)) using a bilayer *h*-BN as the insulating spacer (tunnel barrier) and systematically studied its transport characteristics. Our results demonstrate clear Josephson transport properties in the NbSe<sub>2</sub>/*h*-BN/Nb junction device. We further measured the critical current fluctuations with a 1/*f* spectral density at low-frequency in the NbSe<sub>2</sub>/*h*-BN/Nb Josephson device. We find that the critical current noise in the *h*-BN based Josephson device is at least a factor of four lower than the 1/*f* noise power of the previously studied aluminum oxide-based junctions [10].

#### 2. Results and discussion

We first measured the temperature dependences of the resistances (R) from different parts of the Josephson junction, including the two superconductors (Nb and NbSe<sub>2</sub>) and the  $NbSe_2/h$ -BN/Nb junction itself (figure 1(d)), respectively. As shown in figure 2(a), all three R vs T curves show a sharp transition from a normal resistive state to a zero-resistance superconducting state at the critical temperatures  $T_{\rm c}$ 's, indicating a clearly developed superconductivity. The extracted critical temperatures  $T_c$ 's of Nb, NbSe<sub>2</sub>, and NbSe<sub>2</sub>/*h*-BN/Nb junction (figure 2(a)) are  $\approx$ 7 K, 6 K, and 5 K, respectively. We note that the critical temperatures of Nb (7 K) and NbSe<sub>2</sub> (6 K) in our device are slightly lower than their bulk values (9.25 K for Nb and 7.2 K for NbSe<sub>2</sub>), but higher than the critical temperature  $T_c$  ( $\approx$ 5 K) of the NbSe<sub>2</sub>/*h*-BN/Nb Josephson junction. Our temperature dependences of the Josephson junction device exhibit the direct signatures of superconductivity and Josephson coupling in the NbSe<sub>2</sub>/h-BN/Nb junction through the *h*-BN tunnel barrier. We further note that, compared to other tunnel devices using atomically thin h-BN layers [16, 27], the normal resistance of the NbSe<sub>2</sub>/h-BN/Nb junction is relatively small due to the possible pinholes or defects created in the *h*-BN spacer during the sputtering procedure.

We further studied the current–voltage (I-V) characteristics of the NbSe<sub>2</sub>/*h*-BN/Nb Josephson junction at T = 300 mKas shown in figure 2(b). The I-V curves not only exhibit a hysteresis loop as the bias current is swept back and forth, but also show a clear zero-voltage state when the bias current is less than the critical current  $I_c$ . As the bias current Iis increased above the critic current  $I_c$ , a finite voltage can be measured, indicating that the junction transitions to its normal resistive states, showing the typical behavior of an underdamped Josephson junction. From figure 2(b), we can further extract the Josephson critical current  $(I_c)$  of this device to be  $\approx 120 \ \mu A$  at  $T = 300 \ m K$ . The corresponding critical current density is 2800 A cm<sup>-2</sup> with a junction area of 4.25  $\mu$ m<sup>2</sup>. In addition to the hysteresis around the critical current of  $I_{\rm c} = 120 \ \mu \text{A}$ , we also observed additional voltage jumps around I = 140 and 175  $\mu A$  as shown in figure 2(b). We attribute these features to breaking of superconductivity (at such high currents) in the NbSe2 and Nb layers rather than the NbSe<sub>2</sub>/h-BN/Nb junction itself. All of our I-V results demonstrate a clear Josephson coupling in our NbSe<sub>2</sub>/h-BN/Nb junction device through the atomically thin h-BN spacer,

(a)



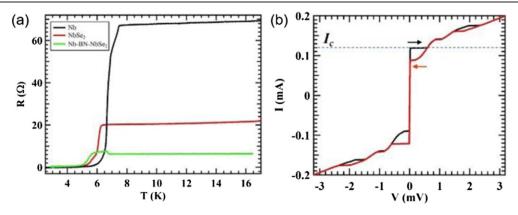
**Figure 1.** A vertical NbSe<sub>2</sub>/*h*-BN/Nb Josephson junction. (a) Schematic illustration of a vertical NbSe<sub>2</sub>/*h*-BN/Nb Josephson junction using a bilayer *h*-BN as the spacer (tunnel barrier). (b)–(d) Device fabrication process of the vertical NbSe<sub>2</sub>/*h*-BN/Nb Josephson junction. (b) A NbSe<sub>2</sub> thin flake with thickness of 15 nm exfoliated on a Si/SiO<sub>2</sub> substrate. (c) Four Ti/Au contacts fabricated on the NbSe<sub>2</sub> thin flake. (d) The final device of the NbSe<sub>2</sub>/*h*-BN/Nb Josephson junction after dry-transferring a thin *h*-BN layer on top of the NbSe<sub>2</sub> thin flake and followed by sputtering a Nb top electrode, with a junction area of 4.25  $\mu$ m<sup>2</sup>. The white dashed line highlights the perimeter of the *h*-BN thin flake. The thickness of the *h*-BN flake is  $\approx$ 1 nm.

which introduces a discontinuity in the superconducting order parameter in the Josephson effect.

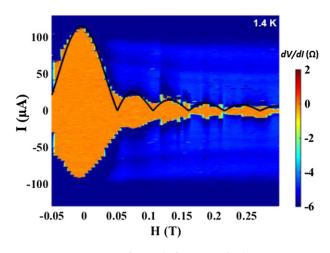
We also examined another hallmark of the Josephson effect, the Fraunhofer effect, in the NbSe<sub>2</sub>/h-BN/Nb junction. It is known that the application of a magnetic field in the junction plane of a Josephson junction induces a gradient in the phase difference between the superconductors, thereby inducing a variation in the critical current  $I_c$  (which is driven by the phase difference). In our measurements, an in-plane magnetic field was applied along the long diagonal direction of the bottom NbSe<sub>2</sub> layer (figure 1(b)). The in-plane magnetic field dependence of the I-V curve measured at T = 1.4 K is shown in figure 3. The critical current  $I_c$  exhibits an approximately periodical modulation by the applied magnetic field. We further extracted a periodicity  $H_0 \approx 50$  mT from the Fraunhoferlike diffraction pattern (figure 3). Theoretically, based on the device geometry of our device, we can estimate the expected periodicity  $H_0 (=\Phi_0/[W(d_t + \lambda_L + \lambda'_L)]) = 13.5$  mT, which is the magnetic field corresponding to a flux quantum  $\Phi_0 =$ h/2e threading the junction cross section area  $A = W(d_t + \lambda_L)$  $+ \lambda'_{\rm L}$ ) = 0.153  $\mu {\rm m}^2$ . Here, W = 3.4  $\mu {\rm m}$  is defined by the width of the bottom electrode (NbSe<sub>2</sub>),  $d_t = 1$  nm is the thickness of the h-BN, and  $\lambda_{\rm L}=39$  nm and  $\lambda'_{\rm L}=5$  nm are the London penetration depths (in the direction perpendicular to the sample surface) of bulk Nb [28] and NbSe<sub>2</sub> [29], respectively. We find that the measured periodicity  $H_0$  ( $\approx 50$  mT) from the Josephson effect is  $\approx$ 3.7 times larger than the theoretically calculated periodicity based on our device geometry and London penetration depths estimated from the bulk values. We speculate that the extended periodicity  $H_0$  observed in our NbSe<sub>2</sub>/h-BN/Nb device can be caused by several factors: for instance, (1) the effective width of the Josephson junction can be shorter than the sample width (3.4  $\mu$ m); (2) the London penetration depths of the two superconductors are highly dependent on their growth conditions and can be significantly different from the reported bulk values that we used in our estimations. If the width of the junction is fixed to 3.4  $\mu$ m, the sum of London penetration depths  $(\lambda_{\rm L} + \lambda'_{\rm L})$  of the two superconductors based on the actual  $H_0$  is expected to be  $\approx 11$  nm only. We further find that the observed  $I_c$  modulation can be fitted to an expected expression of the Josephson junction Fraunhofer pattern  $I_c(H) = I_c(0) \left| \frac{\sin \frac{\pi H}{H_0}}{\frac{\pi H}{H_0}} \right|$ , where  $I_c(0) = 120 \ \mu$ A is the critical current at zero magnetic field. As shown in figure 3, this fitting curve matches reasonably well with the contour of the measured pattern. In addition, the observed single-slit interference-like dependence of  $I_c$  on the magnetic field further confirms that the supercurrent of our NbSe<sub>2</sub>/*h*-BN/Nb junction originates from the Josephson effect through the *h*-BN tunnel barrier, rather than any direct shorting between the Nb and NbSe<sub>2</sub> superconductors.

In a conventional S/I/S Josephson junction, fluctuating conduction channels cause the  $I_c$  to fluctuate with a 1/f power spectrum  $S_{I_c} = S_{I_c}^* (1 \text{ Hz}) / f$ . It has been suggested that the critical current noise is dominated by microscopic defects in the amorphous insulating barrier of the junction or at the disordered metal-insulator interface. In a superconducting qubit, any fluctuation in  $I_c$  modulates the energy level separation between the qubit 0 and 1 states; therefore low-frequency  $I_c$ noise is a potential source of qubit dephasing. These fluctuations also limit the sensitivity of Josephson devices such as SQUID magnetometers. Previous studies have found critical current noise  $S_{I_c}^* (1 \text{ Hz}) = (10^{-6}I_c)^2$  for 100  $\mu$ m<sup>2</sup> conventional (AlO<sub>x</sub> based) junctions at 4 K; moreover,  $S_{I_0}^* (1 \text{ Hz}) / I_c^2$ scales inversely with junction area and as the square of temperature [10–12].

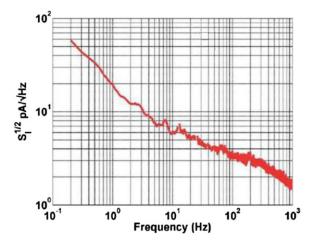
We have performed the  $I_c$  noise characterization on the NbSe/*h*-BN/Nb Josephson junction at T = 3 K. In the experiment, the NbSe/h-BN/Nb junction was voltage biased and a separate dc SQUID circuit was used to monitor the current through the junction. The representative noise spectrum is presented in figure 4. We measured a current noise amplitude at 1 Hz of 20 pA Hz<sup>-1/2</sup> A<sup>-1</sup> in the NbSe<sub>2</sub>/*h*-BN/Nb Josephson junction with a lateral junction area of 4.25  $\mu$ m<sup>2</sup>. Separate tests reveal that this noise is dominated by the noise of the measurement system and, thus, this result represents only an upper bound on the critical current noise of the junction. We note that, such an upper limit, though, is already a factor of 4 lower than that in the conventional oxide-based Josephson junctions (after proper scaling of junction's critical current, area, and temperature) [10-12]. Thus, our result demonstrates the strong promise of h-BN as an improved tunnel barrier for Josephson quantum devices and qubit applications. We note that, in order to circumvent the noise contribution of the readout circuit and access the true intrinsic noise of the *h*-BN junction, one may further decrease the area of the h-BN junction in order to enhance the magnitude of fluctuating conductance channels relative to the overall junction conductance. We further note that in our current devices, defects or pinholes may be created in the h-BN during Nb sputtering. It is expected that the defects and the noise of the h-BN-based Josephson junction can be further reduced by replacing the Nb contact with a 2D superconductor layer to form an all-vdW Josephson junction, where clean interfaces in the Josephson junctions may be achieved.



**Figure 2.** Electrical characterization of superconductivity in the NbSe<sub>2</sub>/*h*-BN/Nb Josephson junction: (a) resistance (*R*) vs temperature (*T*) measured on Nb, NbSe<sub>2</sub>, and the NbSe<sub>2</sub>/*h*-BN/Nb junction, respectively. (b) Current–voltage (*I*–*V*) curves of the NbSe<sub>2</sub>/*h*-BN/Nb Josephson junction measured by sweeping the bias current at T = 300 mK, featuring a superconducting critical current  $I_c = 120 \mu$ A. Arrows indicate the sweep directions of the current.



**Figure 3.** Measurement of Fraunhofer pattern in the NbSe<sub>2</sub>/*h*-BN/Nb Josephson junction. Color plot of dV/dI as a function of the bias current *I* and in-plane magnetic fields at T = 1.4 K, showing a characteristic Fraunhofer-like diffraction pattern with fitting (solid line).



**Figure 4.** The critical current noise spectra of the NbSe<sub>2</sub>/*h*-BN/Nb Josephson junction measured at T = 3 K. The system noise floor is dominated by the noise of the SQUID-based measurement circuit used to monitor the critical current fluctuations and thus represents an upper limit on the actual critical current noise of the *h*-BN-based junction.

#### 3. Conclusion

In conclusion, we have fabricated a vertical S/I/S Josephson junction device using a single-crystalline *h*-BN layer as the insulating spacer between two superconductors (NbSe<sub>2</sub> and Nb). Through the transport studies, we have observed well developed Josephson effect in the NbSe<sub>2</sub>/*h*-BN/Nb junction device. We further demonstrated that, by using the single crystalline *h*-BN in the S/I/S Josephson junction, the critical current noise of the device is at least a factor of 4 lower than the 1/f noise power of the previously studied aluminum oxide-based Josephson junctions. Our results pave a possible way to reduce the noise and improve the sensitivity of SQUID magnetometers, as well as to extend the coherence time of Josephson junctions.

#### 4. Methods

#### 4.1. Device fabrication

We fabricated the NbSe<sub>2</sub>/h-BN/Nb Josephson junctions incorporating a single-crystalline h-BN spacer and two superconductors (NbSe<sub>2</sub>, which is a 2D/vdW superconductor, and Nb). The NbSe<sub>2</sub> thin flakes (10 nm to 30 nm) were mechanically exfoliated from a 2H-NbSe<sub>2</sub> bulk crystal ( $T_c \approx 7.2$  K) and transferred on a SiO<sub>2</sub> (300 nm)/Si (heavily doped) substrate (figure 1(b)). To prevent surface oxidation of the NbSe<sub>2</sub> layer, a layer of polymethyl methacrylate (PMMA) e-beam resist was immediately spin-coated on the substrate. Then, a few windows were opened in the PMMA layer using the standard ebeam lithography. Four Ti/Au (10 nm/60 nm) metal contacts (figure 1(c)) were deposited on the NbSe<sub>2</sub> flake using an ebeam evaporator. To prepare the h-BN tunnel barrier (typically  $\approx$ 1–2 nm-thick, characterized by atomic force microscopy and optical microscopy), we exfoliated the atomically thin layer from the h-BN bulk crystals (HQ graphene) on another SiO<sub>2</sub> (100 nm)/Si substrate. After the lift-off step of the NbSe<sub>2</sub> device fabrication, the selected h-BN flake was transferred on top of the NbSe<sub>2</sub> flake using a dry-transfer technique (figures 1(c) and (d)) [30]. To minimize the possible surface oxidation of the NbSe<sub>2</sub> layer, the whole transfer process was limited to 30 min. Finally, the top superconductor Nb (80 nm thick) was made by another round of e-beam lithography and followed by sputter deposition. The final NbSe<sub>2</sub>/*h*-BN/Nb Josephson device is shown in figure 1(d).

#### 4.2. Device characterization

To study the transport characteristics of the NbSe<sub>2</sub>/h-BN/Nb Josephson junction, the device (NbSe<sub>2</sub>  $\approx$  15 nm; Nb  $\approx$  80 nm) was cooled in a variable temperature insert with a base temperature of T = 1.5 K or a He-3 cryostat with a base temperature of T = 300 mK. All the resistance measurements were using a typical four-terminal configuration. We characterized the temperature dependences of the resistances from different parts of the NbSe<sub>2</sub>/h-BN/Nb Josephson junction. We further measured the I-V characteristics of the Josephson junction. The measurement circuit lines used for the I-V characterizations were electrically filtered by two-stage low-pass RC filters, with a cutoff frequency of  $\approx 10$  kHz, in combination with another set of RC filters and  $\pi$ -type low-pass LC filters with a cutoff frequency of 10 MHz at room temperature. To further measure the critical current noise, the devices were cooled to 3 K using a pulse tube cooler (which is part of an adiabatic demagnetization refrigerator system). The NbSe<sub>2</sub>/h-BN/Nb Josephson junction was voltage biased and fluctuations in the critical current were measured by an auxiliary Nb-AlO<sub>x</sub>-Nb SQUID [31]. Here, the integrated input coil of the readout SQUID was wired in series with the NbSe<sub>2</sub>/h-BN/Nb junction and the readout SQUID was operated in a flux-locked loop with flux modulation at 100 kHz. The noise floor was dominated by the added noise of the measurement system, so we are only able to set an upper bound on the critical current noise of the Josephson junction.

#### Acknowledgments

We acknowledge valuable discussions with C C Yu and experimental help from L Rokhinson. We acknowledge partial support during various stages of the work from DARPA MESO program (Award N66001-11-1-4107), NSF (Award EFMA-1641101), and JSPS Kakenhi (18H03858). JT also acknowledges DOE, Office of Basic Energy Sciences, Division of Materials Sciences and Engineering for financial support under Award DE-SC0021281. YPC also acknowledges support from DOE, Office of Science through the Quantum Science Center (QSC), a National Quantum Information Science Research Center. JT and YPC acknowledge support from the U.S. Department of Commerce, National Institute of Standards and Technology under the financial assistance award 70NANB12H184. The authors thank C-I Liu, G Fitzpatrick, A L Levy, E C Benck, and the NIST Editorial Review Board for assistance with the internal NIST review process. Commercial equipment, instruments, and materials are identified in this paper in order to specify the experimental procedure adequately. Such identification is not intended to imply recommendation or endorsement by the National Institute of Standards and Technology or the United States government, nor is it intended to imply that the materials or equipment identified are necessarily the best available for the purpose.

#### Data availability statement

The data that support the findings of this study are available upon reasonable request from the authors.

#### **ORCID iDs**

Jifa Tian <sup>D</sup> https://orcid.org/0000-0003-2921-470X Albert F Rigosi <sup>D</sup> https://orcid.org/0000-0002-8189-3829 R McDermott <sup>D</sup> https://orcid.org/0000-0001-5677-8637 Yong P Chen <sup>D</sup> https://orcid.org/0000-0002-7356-4179

#### References

- Makhlin Y, Schön G and Shnirman A 2001 Rev. Mod. Phys. 73 357–400
- [2] Clarke J and Wilhelm F K 2008 *Nature* **453** 1031
- [3] Schulze H, Behr R, Müller F and Niemeyer J 1998 Appl. Phys. Lett. 73 996–8
- [4] van Harlingen D J, Robertson T L, Plourde B L T, Reichardt P A, Crane T A and Clarke J 2004 Phys. Rev. B 70 064517
- [5] Ansari M H and Wilhelm F K 2011 *Phys. Rev.* B **84** 235102
- [6] Xi X, Zhao L, Wang Z, Berger H, Forró L, Shan J and Mak K F 2015 Nat. Nanotechnol. 10 765
- [7] McDermott R 2009 IEEE Trans. Appl. Supercond. 19 2–13
- [8] Oliver W D and Welander P B 2013 *MRS Bull.* 38 816–25
  [9] Zeng L J, Nik S, Greibe T, Krantz P, Wilson C M, Delsing P and
- Olsson E 2015 J. Phys. D: Appl. Phys. 48 395308 [10] Nugroho C D, Orlyanchik V and van Harlingen D J 2013 Appl.
- Phys. Lett. 102 142602
- [11] Mück M, Korn M, Mugford C G A, Kycia J B and Clarke J 2005 Appl. Phys. Lett. 86 012510
- [12] Eroms J, van Schaarenburg L C, Driessen E F C, Plantenberg J H, Huizinga C M, Schouten R N, Verbruggen A H, Harmans C J P M and Mooij J E 2006 Appl. Phys. Lett. 89 122516
- [13] Novoselov K S, Geim A K, Morozov S V, Jiang D, Zhang Y, Dubonos S V, Grigorieva I V and Firsov A A 2004 Science 306 666–9
- [14] Novoselov K S, Mishchenko A, Carvalho A and Castro Neto A H 2016 Science 353 aac9439
- [15] Dean C R et al 2010 Nat. Nanotechnol. 5 722
- [16] Britnell L et al 2012 Nano Lett. 12 1707-10
- [17] Yabuki N, Moriya R, Arai M, Sata Y, Morikawa S, Masubuchi S and Machida T 2016 Nat. Commun. 7 10616
- [18] Radisavljevic B, Radenovic A, Brivio J, Giacometti V and Kis A 2011 Nat. Nanotechnol. 6 147–50
- [19] Lee K-H et al 2019 Nano Lett. 19 8287-93
- [20] Li L, Yu Y, Ye G J, Ge Q, Ou X, Wu H, Feng D, Chen X H and Zhang Y 2014 Nat. Nanotechnol. 9 372–7
- [21] Chen W et al 2019 Supercond. Sci. Technol. 32 115005
- [22] Idzuchi H *et al* 2020 arXiv:2012.14969
- [23] Kang K, Jiang S, Berger H, Watanabe K, Taniguchi T, Forró L, Shan J and Mak K F 2021 arXiv:2101.01327
- [24] Ai L et al 2021 arXiv:2101.04323
- [25] Kamalakar M V, Dankert A, Bergsten J, Ive T and Dash S P 2014 Sci. Rep. 4 6146
- [26] Fu W, Makk P, Maurand R, Bräuninger M and Schönenberger C 2014 J. Appl. Phys. 116 074306
- [27] Dankert A, Venkata Kamalakar M, Wajid A, Patel R S and Dash S P 2015 Nano Res. 8 1357–64

- [28] Maxfield B W and McLean W L 1965 Phys. Rev. 139 A1515-22
- [29] Fletcher J D, Carrington A, Diener P, Rodière P, Brison J P, Prozorov R, Olheiser T and Giannetta R W 2007 Phys. Rev.
- [30] Andres C-G, Michele B, Rianda M, Vibhor S, Laurens J, van der Zant H S J and Gary A S 2014 2D Mater. 1 011002
- Rev. [31] Sendelbach S, Hover D, Kittel A, Mück M, Martinis J M and McDermott R 2008 Phys. Rev. Lett. 100 227006
- Prozorov R, Olheiser T and Giannetta R W 2007 Phys. Rev Lett. 98 057003