

# Topological Insulators $\text{Bi}_2\text{Te}_3$ and $\text{Bi}_2\text{Se}_3$ Grown by MBE on (001) GaAs Substrates

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**Abstract.**  $\text{Bi}_2\text{Te}_3$ ,  $\text{Bi}_2\text{Se}_3$  and their alloy films were successfully grown by molecular beam epitaxy (MBE) on (001) GaAs substrates. The structural properties of these films were investigated by Reflection high-energy electron diffraction (RHEED), Atomic force microscopy (AFM), X-ray diffraction (XRD), High-resolution transmission electron microscopy (HRTEM) and Raman spectroscopy and mapping. The results indicate that the epitaxial films are highly uniform. High-field and low-temperature magneto-transport measurements on these films are carried out and discussed.

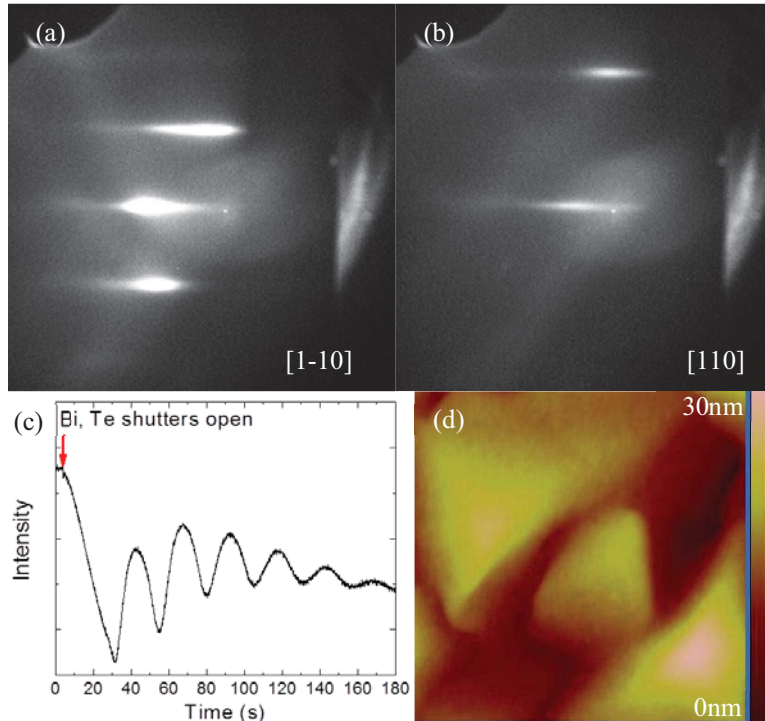
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Recently, photoemission measurements of the surface of topological insulators such as  $\text{Bi}_2\text{Te}_3$  and  $\text{Bi}_2\text{Se}_3$  confirmed that there exists a conducting surface state in these materials with a single Dirac point [1,2]. Although the growth of such topological insulators films by molecular beam epitaxy (MBE) is especially attractive, due to the pseudo-hexagonal structure of  $\text{Bi}_2\text{Te}_3$  and  $\text{Bi}_2\text{Se}_3$ , so far most efforts to fabricate these films by MBE have been carried out using Si (111) [3,4] as substrates, with little work done on GaAs (111) substrates [5]. Because spintronic materials such as GaMnAs are usually grown on (001) GaAs substrates, in this work we discuss MBE growth of  $\text{Bi}_2\text{Te}_3$ ,  $\text{Bi}_2\text{Se}_3$  and their alloys on (001) GaAs substrates, which may open interesting opportunities for applications in spintronics.

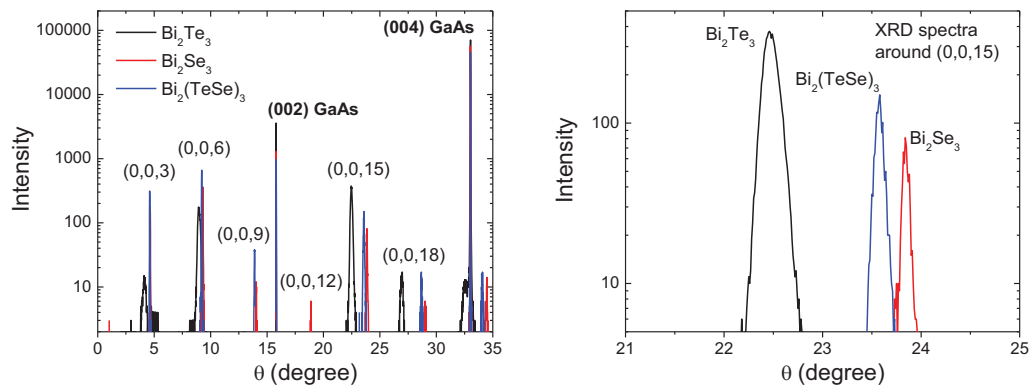
$\text{Bi}_2\text{Te}_3$ ,  $\text{Bi}_2\text{Se}_3$  and their alloy films were grown using a dual-chamber Riber 32 MBE system. First, a (001) GaAs substrate was heated up to 600°C for removing oxidation. The growth was initiated by the monolayer of Te-Bi-Te-Bi-Te or Se-Bi-Se-Bi-Se – a quintuple layer (QL) – in serial atomic layer epitaxy (ALE) type fashion deposited at room temperature. The substrate was then gradually heated to 300°C, and a nice streaky reflection high-energy electron diffraction (RHEED) pattern appeared (see Figs. 1a and 1b). The different RHEED patterns observed in figure confirm the hexagonal surface lattice of  $\text{Bi}_2\text{Te}_3$  and  $\text{Bi}_2\text{Se}_3$  and indicate the  $c$ -axis growth of films. The MBE growth of  $\text{Bi}_2\text{Te}_3$ ,  $\text{Bi}_2\text{Se}_3$  and their alloys was then performed under Te or

Se rich conditions with  $T_{substrate} = 300^{\circ}\text{C}$ . The film growth rate ( $\sim 2\text{nm}/\text{min}$ ) was determined by RHEED oscillations (as shown in Fig. 1c), which agrees with the thickness determined by X-ray reflectivity spectra (not shown).



**FIGURE 1.** (a,b) RHEED diffraction patterns observed in two specific orientations of a (001) GaAs substrate during the growth of a  $\text{Bi}_2\text{Te}_3$  film. (c) RHEED oscillations observed during the beginning of the growth of a  $\text{Bi}_2\text{Te}_3$  film. (d) AFM height image of 136nm-thick  $\text{Bi}_2\text{Te}_3$  film. The size is  $1 \times 1 \mu\text{m}$ .

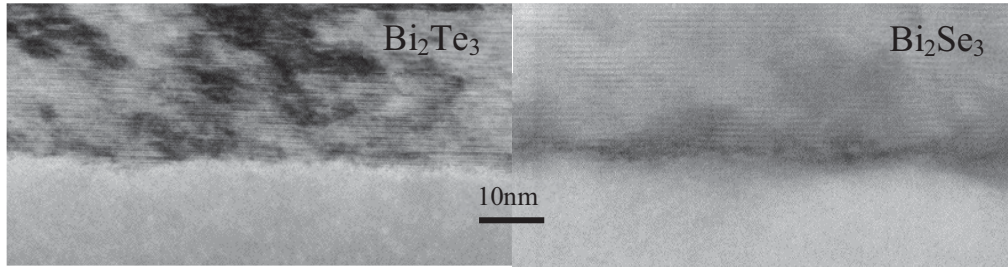
Atomic force microscopy (AFM) measurements (see Fig. 1d) show a surface roughness of  $\sim 5.27\text{nm}$  for a 136nm-thick  $\text{Bi}_2\text{Te}_3$  film. The surface roughness increases as the film thickness increases, suggesting that growth conditions still need to be optimized in the future.



**FIGURE 2.** X-ray diffraction patterns obtained from 233-nm-thick  $\text{Bi}_2\text{Te}_3$ , 220-nm-thick  $\text{Bi}_2(\text{TeSe})_3$  and 180-nm-thick  $\text{Bi}_2\text{Se}_3$  films grown by MBE on GaAs (001) substrates.

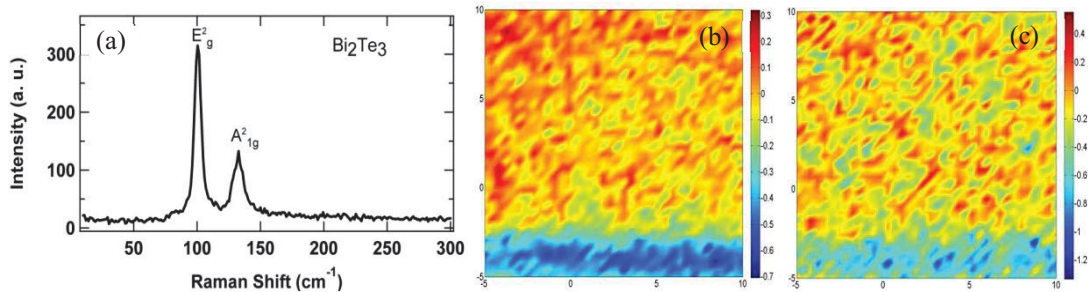
The crystalline structure of the films was confirmed by high resolution X-ray diffraction (XRD) and by transmission electron microscopy (TEM). XRD spectra shown in Fig. 2 reveal reflections only from  $\{003\}$ -type lattice planes, which is indicative of highly pronounced  $c$ -axis growth of the  $\text{Bi}_2\text{Te}_3$  and  $\text{Bi}_2\text{Se}_3$  films. Note that,  $\text{Bi}_2\text{Te}_3$  and  $\text{Bi}_2\text{Se}_3$  have a rhombohedral layered structure, which is composed of the stacking order of hexagonal Te(Se) and Bi atomic layers along the  $z$ -direction. The highly parallel quintuple layers – Te(Se)-Bi-Te(Se)-Bi-Te(Se) – are seen in both  $\text{Bi}_2\text{Te}_3$  and  $\text{Bi}_2\text{Se}_3$  films despite the slightly wavy interface of the GaAs substrate (see Fig. 3). Note that, the light and dark areas visible in the micrographs are possibly due to ion-milling damage during TEM sample preparation.

Raman spectroscopy and Raman mapping have also been performed on these materials using a 532 nm excitation laser (power  $\sim 0.8\text{mW}$ ). The Raman spectra show three characteristic peaks for  $\text{Bi}_2\text{Se}_3$  [at  $\sim 71\text{cm}^{-1}$  ( $A^1_{1g}$ ),  $131\text{cm}^{-1}$  ( $E^2_g$ ) and  $174\text{cm}^{-1}$  ( $A^2_{1g}$ )] (not shown), and two peaks for  $\text{Bi}_2\text{Te}_3$  [at  $\sim 102\text{cm}^{-1}$  ( $E^2_g$ ) and  $134\text{cm}^{-1}$  ( $A^2_{1g}$ )] (see Fig. 4a). The observed peaks are consistent with the lattice vibration modes reported earlier for these materials [6]. The Raman maps show that the positions of the Raman peaks measured within a scan area of  $15\mu\text{m}\times 15\mu\text{m}$  differ by less than  $\sim 1\text{cm}^{-1}$ , indicating a high uniformity of the films, as shown in Figs. 4b and 4c.



**FIGURE 3.** TEM images of  $\text{Bi}_2\text{Te}_3$  and  $\text{Bi}_2\text{Se}_3$  films.

Room temperature electron transport measurements show that the resistivity of the films strongly depends on the Group-VI/Bi flux ratio, thus suggesting a strategy for optimizing future growth conditions. Low-frequency noise measurements show  $1/f$ -type voltage fluctuations, similar to the noise behavior occurring in conventional semiconductor films.



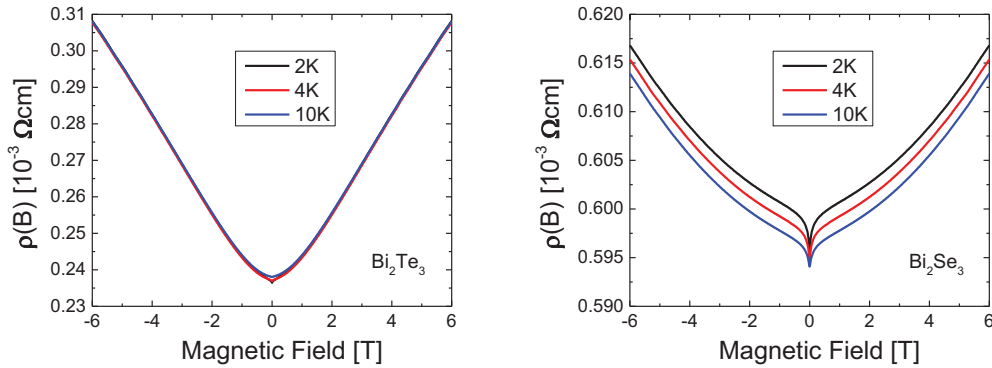
**FIGURE 4.** (a) Raman spectra measured in thin  $\text{Bi}_2\text{Te}_3$  film. (b,c) Raman mappings of peak position differences compared to a bulk single crystal  $\text{Bi}_2\text{Te}_3$  sample for the peaks  $E^2_g$  and  $A^2_{1g}$ , respectively.

High-field and low-temperature magneto-transport measurements on these films were carried out at temperatures ranging from 2K to 10K with fields up to 6T. The results confirm that the undoped topological insulator films have  $n$ -type conductivity,

with carrier concentrations in the range of  $4 \times 10^{19}$  to  $8 \times 10^{19} \text{ cm}^{-3}$  and mobilities in the range of  $270$  to  $330 \text{ cm}^2(\text{Vs})^{-1}$ . The magnetic field dependences of the resistivity with magnetic field applied normal to the plane all show a sharp positive magnetoresistance cusp at low fields and low temperatures, which may be related to weak antilocalization corrections in a 2D system. The cusp is enhanced in thinner films, suggesting that it is a surface-related phenomenon. Our results are consistent with those reported for films grown on other substrates [5,7].

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**FIGURE 5.** The magnetic field dependence of the resistivity for the 20-QL ultrathin  $\text{Bi}_2\text{Te}_3$  and  $\text{Bi}_2\text{Se}_3$  films with the field applied in the out-of-plane direction at various temperatures.

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